

Date: <u>7/20/99</u>	Express Mail Label No. <u>EL29099211245</u>
----------------------	---

Inventors: Yi-Qun Li, Robert C. O'Handley, Gerald F. Dionne
and Chun Zhang

Attorney's Docket No.: SPIN98-01pA

PASSIVE SOLID-STATE MAGNETIC FIELD SENSORS
AND APPLICATIONS THEREFOR

RELATED APPLICATION

This application claims the benefit of U.S. Provisional Application No.
5 60/094,837, filed July 31, 1998, the contents of which are incorporated herein by
reference in their entirety.

BACKGROUND OF THE INVENTION

Technical field

This invention is in the field of magnetic sensors which detect a magnetic field
10 or magnetic field change and find use in linear or rotary motion detection, electrical
current sensing, linear or rotary position sensing, magnetic imaging, magnetic recording
read heads, magnetic recording media as well as general magnetic field sensing.

Background of the Invention

Because of the many applications of magnetic field sensors, there is a long
15 history of technical development of materials and means to measure fields of various
strengths. No one sensor can perform every function well. Factors such as size, weight,
power consumption, and cost should be minimized by a field sensor. Sensitivity,
linearity, bias, stability, reliability, and operating temperature and frequency range are
factors that should be optimized. As with any instrument, it is usually difficult to

achieve all of these characteristics in one device. The most common magnetic sensors used in a variety of applications are the Hall effect sensor and the variable reluctance coil. The drawback of variable reluctance devices is that they generate signals proportional in size to the time rate of change of magnetic flux. The signal size
5 therefore decreases with decreasing speed, and below a certain flux change rate, the signal disappears into the noise. Hall effect devices generate a very small raw signal because of low field sensitivities ($0.5 \sim 5 \text{ mV}/100 \text{ Oe}$ applied field). This mandates signal conditioning, and requires that a certain minimum field be available for device operation.

10 The concept in combining the magnetostrictive materials and piezoelectric layers for highly sensitive magnetometer was first introduced by Mars D. Mermelstein in 1986. In his patents US 4,769,599 and US 5,130,654 a magnetometer was disclosed as a device using piezoelectric resonator to create a standing stress wave in the sensing magnetostrictive ribbon and using pickup coil to readout the electromotive force. A
15 minimum detectable field gradient of $7.7 \text{ pT}/\text{cm Hz}$ was achieved in this device by using a differential amplifier technique. In October 1997, a device called piezomagnetometer was patented by Walter N. Pondney. In that device a stack of 201 alternating piezoelectric and magnetostrictive layers was used, in which 100 pairs of piezoelectric-magnetostrictive layer capacitors are connected electrically in parallel to
20 increase the charge storage by raising the effective capacitor plate area. The device requires multiple layers that are placed in a permanent biasing magnetic field normal to the layer surface. The resolution as high as $1 \text{ pT}/\text{cm Hz}$ was achieved.

SUMMARY OF THE INVENTION

Our invention represents a new technology of passive solid-state magnetic
25 sensors based on a combination of magnetostrictive material and piezoelectric material. The innovative sensors can be mass produced at low cost in comparison with any existing magnetic sensor technologies including variable reluctance coils, hall-effect

devices, magnetoresistance semiconductors, and the most recently developed giant magnetoresistance (GMR) metal multilayers. Such a magnetostrictive/piezoelectric sensor does not consume any electrical power and easily has a field sensitivity larger than 10mV/Oe. The devices including digital speed meter, digital flow meter, and
5 digital electrical current meter have been demonstrated as application examples of the inventive sensors.

The combination of piezoelectric and magnetostrictive layers has not yet been used for the general applications of magnetic sensors. The present invention is basically a device that has two magnetostrictive layers and one piezoelectric layer or two
10 piezoelectric layers and one magnetostrictive layer. Three-layer device can be connected electrically in series to increase the voltage by raising capacitor plate separation. The sensors are designed for general magnetic field detection purposes in applications ranging from speed, flow, and electrical current detection to the information storage and imaging. The advantages of the inventive sensors over the
15 competitive technologies are passive solid-state, high field sensitivity, wide dynamic range up to several thousands of Oersted, and low-cost in manufacturing.

Generally, the present invention uses a piece of piezoelectric material in contact with a magnetostrictive material. The magnetostrictive component strains in response to a magnetic field. This strain couples to the piezoelectric element causing it to
20 produce an electrical output signal. The main objective of the invention is to provide an innovative, low cost and highly sensitive magnetic sensor of the simplest design, which requires no electrical power for the sensing element. The sensitivity of the device and the operating magnetic field range can be adjusted through material properties and structure designs for a variety of applications. The important variables used in the
25 design include 1) selecting magnetostrictive materials with appropriate properties, 2) selecting piezoelectric materials with optimal properties, 3) determining the optimal number capacitive element, 4) selecting the appropriate thickness of each layer, 5)

designing the geometry of the structure, and 6) establishing the most efficient bonding and packaging methods. The inventive magnetic sensors can be widely used in replacement of Hall-effect sensors, variable reluctance coils, and magnetoresistive devices. The particular applications of the inventive magnetic sensors are (a) speed
5 detection and controls for rotary machines including automobiles, airplanes, locomotives, etc., (b) flow meters for reading and controls of liquid or gas flows, (c) electrical current meters for reading and controls of electrical power usage, and (d) micromagnetic field sensors for magnetic recording/reading heads and magnetic imaging devices.

10 BRIEF DESCRIPTION OF THE DRAWINGS

The foregoing and other objects, features and advantages of the invention will be apparent from the following more particular description of preferred embodiments of the invention, as illustrated in the accompanying drawings in which like reference
15 characters refer to the same parts throughout the different views. The drawings are not necessarily to scale, emphasis instead being placed upon illustrating the principles of the invention.

Figs. 1A-1C are schematics of the basic structures of the passive solid state magnetic sensors according to the invention.

Fig. 2 is a schematic of the cantilever structure of the passive solid state
20 magnetic sensor according to the invention.

Figs. 3a and 3b are schematics of planar piezoelectric design with magnetostrictive ferrite substrate according to the invention.

Figs. 4a-4c are open-loop circuit designs for direct measurement of voltage generated by a piezoelectric magnetic sensor; and a closed-loop circuit designs for
25 measurement of the charge change produced by a piezoelectric magnetic sensor according to the invention.

Figs. 5a, 5b, 5c, 5d are schematics showing a piezoelectric magnetic read/write head according to the invention.

Fig. 6 shows a passive piezoelectric magnetic random access memory (P-MRAM) showing simplified view of trilayer magnetic/piezo/magnetic elements at left and conductors at right. The passive voltage measured across the spin valve is depicted at the lower left. A cross section of the main elements of the device is shown at lower right according to the invention.

Fig. 7 is a schematic of two dimensional piezoelectric magnetic sensor array for magnetic imaging according to the invention.

Fig. 8 shows the signal output (upper trace) of a structure-B sensor when an ac magnetic field (lower trace) is applied according to the invention.

Fig. 9 shows the RMS of a signal output from a structure-B sensor as a function of the peak value of an ac magnetic field.

Fig. 10 shows the RMS of a signal output from a structure-B sensor biased at 12 Oe as a function of the peak value of an ac magnetic field.

Fig. 11 RMS of a signal output from a structure-C sensor using $\text{Fe}_{50}\text{Co}_{50}$ as a function of the peak value of an ac magnetic field.

Fig. 12 shows the passive solid-state magnetic sensor used for detecting the rotating speed of a plate with six micro magnets mounted on the edge in equal distance according to the invention.

Fig. 13 is a schematic of a microprocessor-based electrical circuit design for the passive solid-state magnetic sensors used in the motion detection according to the invention.

Fig. 14 is a schematic of a conventional Gear-Tooth method for generating pulse signals in magnetic sensors according to the invention.

Fig. 15 includes schematics of the principles of the passive magnetic sensor used in a conventional Gear-Tooth method for speed detection according to the invention.

Fig. 16 is a schematic of the passive solid-state magnetic sensor used in metering liquid flow rates according to the invention.

Fig. 17 is a schematic of flow sensor designed in use with Swedgelock and PVC fittings according to the invention.

Fig. 18 shows rotor speed measured as a function of liquid flow rate by a piezoelectric magnetic sensor.

Fig. 19 is a schematic of electrical current measurement by a piezoelectric magnetic sensor according to the invention.

5 Fig. 20 shows the highly linear correspondence of piezoelectric signal output to electrical current.

DETAILED DESCRIPTION OF THE INVENTION

We have discovered an efficient way to incorporate a magnetostrictive material with a piezoelectric material to produce a magnetic field sensor and designed a suitable
10 electrical circuit for impedance matching and signal amplifying. The principle of such a sensor is that a strain is induced in a magnetostrictive element when it is exposed to a magnetic field sufficient to change the direction of magnetization. This magnetostriction strain is then transferred into the adjacent piezoelectric element to which it is bonded. The piezoelectric element, in turn produces an electrical signal
15 output.

Single Element With Magnetostrictive Metal

Figs. 1A-1C illustrate embodiments of the invented magnetic sensor. The structure of Fig. 1 A is a basic two-layer sensor containing one piezoelectric layer and one magnetostrictive layer. Structure of Fig. 1b is comprised of two structure-A
20 sensors; here the magnetostrictive layer will strain simultaneously on both sides of piezoelectric layers. Structure of Fig. 1c is also comprised of two structure-A sensors; here the two magnetostrictive layers will strain simultaneously on both sides of a piezoelectric layer. The thickness of individual element and the total number of elements comprising the structure can be optimized for different applications.

25 Structure of Fig. 1c is preferred for most of the applications since this structure is more rugged and more cost effective. In these structures, the electrical signal is

produced by the strain generated at the interfaces of magnetostrictive and piezoelectric layers by a magnetic field without involving the bending of the structure, which provides a good mechanical stability. By expanding the separation for high voltage and reducing surface areas, multiple sensors can be readily connected in planar geometry.

- 5 This concept is particularly appreciated if the magnetostrictive element is an insulator e. g., a ferrite.

The use of cantilevers is a common method of amplifying mechanical motion to enhance the strain induced in a piezoelectric or magnetostrictive material. This is shown in Fig. 2. In this case a bimorph structure of two piezoelectric layers with one
 10 magnetostrictive layer in the middle is preferred and the two piezoelectric layers are poled in orthogonal directions so that the stresses in the two piezoelectric layers due to bending are in opposite directions. Although this cantilever design produces a larger signal, the structure of Fig. 1c is found to have sufficient sensitivity for most of the applications.

- 15 A soft magnetic material is one in which the direction of magnetization M is readily changed by a modest applied field ($H < 100$ Oe). The simplest effect that governs the orientation of magnetization in a soft magnetic material in the absence of an external field is sample shape. The magnetization vector chooses an orientation, which minimizes the number of magnetic poles appearing on the sample surface. Thus M
 20 prefers to lie along the longest sample dimension and avoid orientations normal to the largest area surfaces. This magnetostatic effect can be used to advantage to design a sensor with a preferred quiescent state of magnetization. Thus a piece of soft magnetic material in ribbon, sheet or thin film form will be magnetized in the longer of the in-plane directions. A field transverse to this quiescent state will cause a rotation of the
 25 magnetization into the field direction. A 90° rotation of the magnetization causes the largest shape change in a magnetostrictive material. A field applied along the preferred axis of magnetization will produce only 180° domain wall motion which does not

change the state of strain of the magnetic material. Therefore, from the view of minimizing the demagnetizing field (increasing the effective magnetic field) in the sensor, the sensor is preferred to place with the longer side parallel to the sensing magnetic field, and from the view of maximizing the magnetostriction in the sensor, the sensor is preferred to place with short side parallel to the sensing magnetic field. The trade off has to be made when the sensor is designed.

The preferred direction of magnetization can also be controlled by magnetic field annealing. Annealing in a field of 100 Oe min at temperatures below the Curie temperature of the magnetic materials. The annealing temperature and time, alloy composition and T_C , cooling rate, and the strength of the field all affect the results obtained. The bias magnetic field can be also applied in certain direction to optimize the design.

Two factors have to be considered for selecting piezoelectric materials. The piezoelectric voltage constant g (Vm/N) determines the sensitivity of generating electrical voltage at a certain force or stress. The temperature coefficient of the g constant determines the operation range of the sensors. Two material candidates which possess high g constant are lead zirconate titanate ($PZT = Pb(Zr_xTi_{1-x})O_3$) ceramics and PVDF (polyvinylidene fluoride) polarized polymers. The former has g constants of 10 to 50 and a high electromechanical coupling coefficients K of 0.3 to 0.7. The later has a high g constant of 200 to 300 but low electromechanical coefficient of 0.1 to 0.15. The major difference between these two materials are their dielectric constants. The PZT ceramics has relative dielectric constant ϵ/ϵ_0 in 100 times larger than the PVDF polymer. Since the figure of merit for the inventive sensor is proportional to gK/ϵ , the PVDF polymer has much higher value of figure merit than PZT. The temperature stability of the PVDF, however, is much worse than PZT ceramics. The other piezoelectric material such as aluminum nitride (AlN) and quartz (SiO_2) having similar figure of the merit to PZT can also be used for the inventive sensors. These two

materials are a better choice for multilayer thin film sensors because of their deposition compatibility with the magnetostrictive metals.

The two major requirements on the magnetic material are:

1. A non-zero magnetostriction so that it strains upon being magnetized.
2. An anisotropy that allows the material to be saturated in a field of strength governed by the range of fields to be measured. This involves both the hysteresis properties of the material and its saturation magnetization value which influences anisotropy depending on shape of the material and direction of the magnetizing field.

For measuring very weak fields, a figure of merit is the ratio

$$Q = E\lambda_s^2/2Ku$$

Q is maximized when the material gives the most strain in the weakest fields.

Amorphous magnetic alloys can be field annealed to be greatest magnetized to saturation in fields below 10 Oe. In a material such as a-FeBSi ($\lambda_s \sim 35 \times 10^{-6}$, $M_s \sim 1400$ emu/g₁^v), $E \sim 10^{12}$ d/cm², $K_u = M_s H_a/2 \sim 7 \times 10^3$ erg/cm³, so $Q \sim 0.18$.

For a highly magnetostrictive material such as Terfenol-D, $E \sim 5 \times 10^{11}$ d/cm³, $\lambda_s = 1750 \times 10^{-6}$, $M_s \sim 800$ emu/cm³, $K_a \sim 1 - 4 \times 10^5$ erg/cm³. Thus

$$Q \sim 0.01$$

which is lower than that for the lower λ_s material. However, Terfenol-D will be suitable for measuring larger field strengths.

Nickel and many iron-nickel alloys are good choices because both anisotropy and magnetostriction can be controlled over reasonably wide ranges. However,

amorphous magnetic alloys remain the best choice for low field measurements because very low magnetic anisotropy can be achieved. Iron-cobalt alloy has a relative large magnetostriction and high saturation field. The magnetostriction of the annealed $\text{Fe}_{50}\text{Co}_{50}$ alloy can reach 150×10^{-6} and the maximum $d\lambda/dH$ can be as high as 2×10^{-6} per

- 5 Ostered in comparison with 4×10^{-6} for FeSiB amorphous ribbon and 0.04×10^{-6} for Ni. The advantage of using FeCo alloy is that the linear range and the sensitivity of magnetostriction verses a magnetic field can be controlled by annealing process to meet a variety of applications.

10 B. Planar Piezomagnetic Design with Magnetostrictive Ferrite Substrate

An alternative device structure is based on the use of a magnetostrictive insulator (ferrite). In combination with a piezoelectric, the ferrite can provide an efficient two-layer planar structure that lends itself to thin film processing and has the advantage of a high length to area ratio to give a small capacitance that permits high-frequency

- 15 operation. In Figs. 3a and 3b, a ferrimagnetic material with magnetostrictive properties acts as a substrate for a pattern of parallel piezoelectric stripes of length l . As indicated, the piezoelectric stripes are electrically polarized (poled) into a ferroelectric state with an initial dielectric displacement bias, and are connected in series such that the polarization directions are continuously reinforcing.

- 20 Fig. 3a depicts the case of alternating pole directions and a simple connector scheme.

The arrangement in Fig.3b is more complicated but offers the processing advantage of allowing the ferroelectric poling step to be delayed until after the device structure is completed.

One important benefit of ferrites is their high initial permeability and narrow hysteresis loop, which avoids the necessity of an external bias magnet. Although the magnetostriction constants are more than a factor of ten smaller than metallic materials such a terfenol-D, ferrites such as NiZn spinel ferrite can have initial permeabilities
 5 more than a factor of one hundred greater. The effect of these magnetic parameters on piezomagnetic device performance can be deduced from the following relation for voltage sensitivity to magnetic field (which is analogous to the Q figure of merit):

$$dV/dH = (d/\epsilon)(\lambda_s/H_K)cl \cos\theta ,$$

where c is Young's modulus. The key design parameter for the ferrite is the ratio of
 10 saturation magnetostriction constant λ_s to the magnetocrystalline anisotropy field H_K . In most ferrites, H_K can be effectively canceled by cobalt additions. The key design parameter of the piezoelectric layer is the ratio of the relevant piezoelectric constant d to the corresponding dielectric permittivity ϵ . The key device design parameter for maximum voltage is the length l of the polarized stripe. All of these design parameters
 15 must be maximized for optimum sensitivity.

One further advantage of this planar device configuration is the large dynamic range of magnetic field sensing that can be achieved simply by adjusting the angle θ between the magnetic field vector and the plane of the ferrite substrate. With hysteresis switching times of ferrites in the nanosecond range and the low capacitance resulting
 20 from small length to area ratios, application of these devices can be extended to the high-frequency sensing regime.

The principle of the generation of the electrical signal from the innovative sensor is based on the piezoelectric effect of ferroelectric or paraelectric materials. The sensor
 25 structure can be treated electrically equivalent to a voltage source in series with a capacitor or a charge generator (current source) in parallel with a capacitor.

Fig. 4a-4c shows two three basic circuit approaches for amplifying and reading the signal output from the sensors.

In the case of Fig. 4a the charge change in the sensor is directly measured as an open loop voltage by an operational amplifier with a high impedance and a low leakage
5 current. In the case of Fig. 4b and 4c the charge change in the sensor is measured as a current passing through a resistor in a closed loop circuit.

Circuit of Fig. 4a is preferred for the inventive sensors because the cut-off frequency (3 dB down) can be as low as 8×10^{-5} Hz with 2 nF of sensor capacitance and 0.08 Hz with 2 pF of sensor capacitance if the input resistance of the operational
10 amplifier and the sensor internal resistance are in a order of $10^{12} \Omega$. More importantly, the measured voltage with this circuit is independent of the area of the sensor, which allows a sensor with a miniature size, for example, for magnetic imaging and magnetic recording read heads. A unit gain follower with a picoamper electrometer amplifier AD546 (impedance as high as $10^{12} \Omega$) was built to test sensors made from
15 FeCo/PZT/FeCo three- layer structure. Frequency independence of the signal output after the cutoff frequency is confirmed in a sensor with a size of $2 \times 10 \text{ mm}^2$ and a capacitance of 2nF. The independence on area of the signal output is confirmed with a $2 \times 10 \text{ mm}^2$ sensor and a $1 \times 5 \text{ mm}^2$ sensor. The ratio of sensor width to sensor length was maintained the same to eliminate the difference of diamagnetizing effect. The other
20 characteristics of the circuit of Fig. 4a to measure the output of the sensor is that the output voltage is theoretically proportional to the thickness of the piezoelectric materials in this layered configuration, which has also been demonstrated by a comparison of two sensors made from different thickness of PZT plates, one is 200 μm and the other is 500 μm thick, the later shows the signal output twice larger than the former.

25 The thin-film devices based on a multilayer structure alternating with magnetostrictive layer and piezoelectric layer are important for reducing the sensor size

and integrating with semiconductor IC circuits. Two potential applications are outlined as the following. A trilayer structure as shown in Fig. 1 can be used to read the stored information on the recording media. The thickness of the devices can be designed as small as 300 Å to meet the requirement for the rapidly growing density of recording media. The advantage of these devices over the currently used MR or GMR technology includes high sensitivity, high thermal stability, and no requirement for input power.

Fig. 5a shows the film structure having a central piezoelectric layer sandwiched by two soft magnetic layers of non zero-magnetostriction. The soft layers are processed (by field annealing or intrinsic stress from substrate) so that they are magnetized essentially parallel to the track width of the underlying bits. The thickness of the piezoelectric layer forms the read head gap, g , and must be less than about half a bit length: $g < \lambda/2$.

When the head is over a dibit ($\pm H_z$ separated by $\lambda/2$) the magnetization of one pole piece is tilted down from the horizontal and the other up. The magnetization change in the two layers results in a change in the state of strain of the piezoelectric layer and a voltage is induced in it. The induced voltage is read directly across the two metallic pole pieces. It does not matter whether the magnetization vectors in the two pole pieces are parallel or antiparallel to each other (the latter is more stable magnetostatically). In either case, the field above the recorded transition will result in a characteristic voltage across the piezoelectric layer. This voltage signal will differ in the parallel and antiparallel cases. The two pole pieces of the head can be set to one or the other state, whichever is optimal for the selected piezoelectric material and the characteristics of the recording medium. The ideal magnetic layer composition will have a weak cross track anisotropy and non-zero magnetostriction (e.g. $\text{Ni}_{76}\text{Fe}_{24}$ or $\text{Fe}_{50}\text{Co}_{50}$). The coercivity of these layers should be minimal.

It is also possible to integrate the write head into this read head structure by making one of the read pole pieces of a high flux density material ($\text{Fe}_{50}\text{Co}_{50}$). The piezoelectric layer will still be strained to produce a voltage during the read process. The read/write head must be shielded and one of the shields provides a low reluctance return path for the write process. Three possible constructions for read heads that are integrated with an inductive write head are shown in Fig. 5b. The view in each case is of the “air bearing surface”, i.e. the end of the recording head looking up from the medium along the z direction. In all cases, the piezoelectric layer thickness defines the read gap, g . The insulation between the shields and the other metallic layers is typically alumina.

In configuration (a), the read sensor is fabricated between two much thicker shield layers. The lower shield serves as a return flux path for the write process, P1. The bits are written across the write gap, with one of the read pole pieces being of a high flux density material and also serving as the other write pole piece, P2. The track width, w , is defined by P2. In configuration (b), the lower shield serves as both P1 and one of the read pole pieces. Thus the read and write gaps are both defined by the thickness of the piezoelectric layer. Again P2 defines the track width. In configuration (c), the read and write gaps are again separated.

It is possible also to make use of the benefits of passive piezoelectric sensing of different magnetic states in MRAM-like devices for nonvolatile information storage. We describe one embodiment here.

In a non-volatile MRAM, writable memory cell is a trilayer of soft-magnetic/metallic spacer/semihard magnetic films. The information is written and stored in the semihard layer. The readout is done by sensing the spin valve resistance across the trilayer in two states of magnetization for the soft or free layer. Because the soft and semihard layers have different coercivities, the resistance changes as a function

of field as shown at the lower left of the figure. Fields greater than H_{c2} are required to write information (one of two stable states of magnetization) in the semihard layer. A field of magnitude between H_{c1} and H_{c2} changes the state of magnetization in the free layer without altering the written bit. When the free layer magnetization is changed, the sense of the resistance change across the device indicates the state of magnetization in the storage layer. The fields are applied with a cross pattern of conductors that also leave to locate the address of each bit. While spin-valve based MRAM has many advantages, it does draw current and the signal is proportional to the sense current. It may also be frequency limited by eddy current losses in the leads and switching times in the magnetic layers.

By simply replacing the Cu layer with a piezoelectric layer as shown in Fig. 6, the need for a sense current is removed. When the state of magnetization is the same in the two magnetic layers, the piezoelectric layer will be in one state of strain (depending on the magnetostriction and thickness of the two adjacent layers). When the state of magnetization is different in the two magnetic layers, the piezoelectric layer will be in a different state of strain. These different states of strain will be reflected in the voltage across the piezoelectric layer which we have demonstrated to be appreciable and readily measured. This voltage is read at the x - y address of the bit as in a spin-valve MRAM but there is no need to power the device by putting a current through it. There are significant advantages to this piezoelectric-MRAM (P-MRAM) in terms of speed, signal-to-noise ratio, and power consumption relative to all metal MRAM. In addition, it may also add to the lifetime and reliability of the devices because issues of metallic interdiffusion across two metallic interfaces are obviated by the presence of a stable oxide material. A variety of other soft and semihard magnetic materials which have significant magnetostriction could be used instead of the ones illustrated in the figure above. In addition, there are a great many piezoelectric materials that can be chosen based on material compatibility, signal to noise, reliability and so on. The shunts, word lines and sense lines can also be made of conductors other than Au.

Two dimensional, as schematically shown in Fig. 7 or one dimensional multilayer sensor arrays can be made for directly magnetic field imaging. The developed readout integrated circuits for ultrasonic imaging and pyroelectric IR imaging can be applied here for detection of the electric charge change in piezoelectric layer
 5 caused by a magnetic field change through magnetostrictive layer. The readout circuit will be directly integrated with the magnetic sensor array on a semiconductor wafer. Such a device will be useful for inspection of magnetic materials including recording disks, medical imaging or for high speed recording read head array.

Example 1

- 10 A sample of piezoelectric lead zirconate titanate (PZT) ceramic sheet (2" × 3" × 150 μ m thick) was first cut into one centimeter square pieces. A magnetic amorphous alloy (FeNiMoSiB) ribbon with a thickness of 30 μ m was also cut into one-centimeter square. A sandwich structure B of Fig. 1 was made by gluing a magnetic alloy layer in the middle with two PZT layers outside using conducting silver epoxy. The two PZT
 15 layers were oriented with the same electric pole direction. Two electric leads were soldered to the sandwich, one on the top surface and one on the bottom surface as shown in Fig. 1b . The sample was placed in an ac magnetic field and the two electrical leads were connected to an oscilloscope. Fig. 8 shows a response signal output (upper trace) with peak to peak signal of 90 mV. The ac magnetic field (lower trace) shows
 20 amplitude of 10 Gauss. The double frequency of the output signal with respect to that of the magnetic field is due to the quadratic dependence of magnetostriction on magnetic field: $e \sim \lambda_s (\cos^2 \theta - 1/3) \sim \lambda_s [(H/H_a)^2 - 1/3]$. The RMS output signal measured as a function of the magnetic field is shown in Fig. 8. The linear range of the output signal is from 5 to 15 Gauss and the signal is saturated when the field is over 25
 25 Gauss. The magnetic field sensitivity of the sample over linear range is approximately 5 mV/Oe as calculated from the curve in Fig. 9. The same measurement was performed on the sample in the presence of a DC bias magnetic field of 12 Oe. The signal output

now has the same frequency as the drive frequency of the magnetic field as long as the amplitude of the magnetic field is smaller than the bias magnetic field of 12 Oe. The RMS output signal is plotted as a function of the magnetic field in Fig. 10. A magnetic field sensitivity of 20 mV/Oe is obtained in this case.

5

Example 2

In this example two 3" x 4" x 50 μm $\text{Fe}_{50}\text{Co}_{50}$ sheets (as-rolled) and one 3" x 4" x 175 μm PZT plate were glued together to form the structure C as shown in Fig.1 by a manufacturing process provided by Morgan Matroc, Inc. located at Bedford, Ohio. 200 pieces of 2 mm x 10mm sensors were made by dicing the 3 "x 4 " glued wafer. The sensitivity of the selected samples was tested to show a good uniformity. Using a 1/8" x 1/4" NdFeB magnet placed at one end of the sensor serving as a bias magnetic field, these sensors have the sensitivity of 5 mV RMS signal per Oe and a large linear range as shown in Fig. 11.

10

Example 3

One of the example-2 devices was used to demonstrate the detection of rotation speed. An aluminum plate mounted with six micro NdFeB magnets equally spaced at the edge, as shown in Fig. 12, was driven with a DC motor. A sine-wave magnetic field generated by this rotating aluminum plate was then detected by the sensor with an electrical sine-wave signal output. An electrical circuit, including impedance matching, wave modification, amplifying, pulse counting and displaying, was designed for the sensor as shown in Fig. 13. A square-wave pulse signal was measured from the circuit output. The number of the pulses was then counted within a minute, converted to the speed of the motor rotation, and displayed on the liquid crystal screen by the circuit. A software used in a microprocessor was also developed to calculate the time between the pulses and then to convert it to the frequency or revolution per minute (RPM). This

15

20

25

approach eliminates the necessity of using sixty teeth gear for achieving resolution of one RPM. With this developed software we have built a tachometer which can read the speed from a wheel having only one magnet (one pulse per resolution) with a resolution of one RPM. In this configuration the sensor can be placed as far as 15 mm away from the rotating aluminum plate. We have also designed the sensor for ferromagnetic gears which are commonly used in the industry as shown in Fig. 15. A bias magnetic field was applied by placing a 1 cm-diameter ferrite permanent magnet in contact with the long side of a rectangular example-2 device. A commercial 24 pitch ferromagnetic gear was used. A schematic of magnetic field inside the sensor at tip and at valley of the gear is shown in Fig. 16. Although the sensor has to be placed within 5 mm of the gear, an electrical pulse signal was clearly detected and converted to the speed of the rotation by the same electrical circuit as shown in Fig. 13.

Example 4

One of the example 2-sensors was also demonstrated for the application of measuring liquid flow rate. In this example a commercial water flow indicator was purchased and six micromagnets was placed on the tips of each blade as shown in Fig. 16. An magnetic sensor was attached to the outside of the flow indicator. The water flow rate was then measured as a function of the rotation speed of the blade inside the flow indicator. The calibration and the conversion of the flow rate unit is easily achieved by a microprocessor used in the electrical circuit shown in Fig. 13. We have also designed a flow sensor to fit the popular pipe fittings such as Swedglock Brass Tee and PVC Tee as shown in Fig. 17. A rotor with four miro NdFeB magnets mounted on four blades, respectively, are moving freely at the bottom of the flow sensor. A flow sensor was sealed inside the Tee at the optimized height which gives the maximum rotation speed at a certain flow rate. Fig. 18 shows a linear relationship between rotor speed and the liquid flow rate.

Example 5

One of the example-2 sensors was also demonstrated for measuring ac electrical current. Two bias magnets were placed at each end of the sensor as shown in Fig. 19. The experiment was performed by placing a sensor on top of electrical wire carrying a current from 1 to 10 A, and a raw signal was measured by an oscilloscope. The result plotted in Fig. 20 shows a linear relationship between the RMS signal and the current. The sensitivity is as large as 2 mV/A without amplification. The improvement of one order of magnitude higher sensitivity can be achieved by optimizing the material properties and the geometrical design. The linear fitting curve passing the origin point shown in the figure indicates the characteristic of the non zero-offset voltage for this current sensor in comparison with zero- offset voltage in the case of Hall effect and GMR current sensors. This advantage will lead to a simpler design of precision current sensors. A large dynamic range of the measurement is ready to be achieved by selecting the magnetic materials and designing the sensors.

While this invention has been particularly shown and described with references to preferred embodiments thereof, it will be understood by those skilled in the art that various changes in form and details may be made therein without departing from the spirit and scope of the invention as defined by the appended claims.